



**Overview**

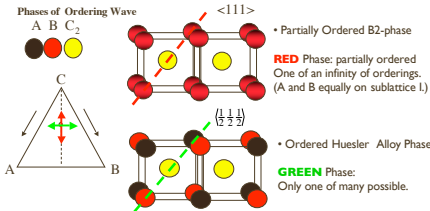
Technologically or scientifically interesting alloys have multiple components and sublattices. As a result, N-component alloys have “complex” ordering with up to (N-1) phase transitions and arrive at the ground-state from an *infinite* number of possible high-temperature, partially-ordered phases.

Predicting the N-1 transitions, their electronic origin, and the short-range order (SRO) and long-range order (LRO) at fixed composition in alloys, is crucial for interpreting experiment and for materials design. Using both *classical* and *electronic density-functional-theory* (DFT) methods [1], a thermodynamic theory of ordering is possible based upon electronic structure and energetics, multiple-scattering (KKR) theory, in particular [2].

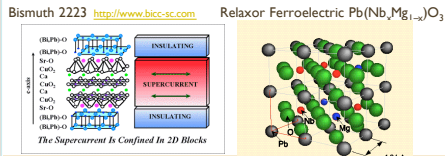
**Why are alloys complex?**

N-component alloys have an *infinity* of choices for ordering [3], e.g., site occupancies in ternary (N=3) bcc ABC<sub>2</sub> alloy with k=(111) SRO peak has N-1 (or 2) phase transitions:

disorder → partially LRO → fully LRO



**Multisublattice examples**



**Goal**

- To improve the electronic DFT configurational averaging in (partially) disordered alloys (originally based on the single-site Coherent Potential Approximation) by including local, multi-site configurational effects in “systematically exact” manner via reciprocal-space *coarse-graining* concepts developed within dynamical mean-field theory [4].
- To extend this DFT-based thermodynamic theory of ordering to general partially-ordered states (i.e., with multi-sublattice orderings) so as to compare directly to the k-space short-range order measured experimentally.
- To improve the mean-field used within the exact classical DFT. In particular, to correct the atomic self-energies by summing **all** *cyclic diagrams* to O(1/Z), where Z is the number of neighbor. These correction maintain required intensity sum rules (violated in most mean-field theories) and renormalize correlation in k-dependent manner.

**Thermodynamic Theory of Ordering from Combined Classical and Electronic DFT**

The thermodynamic average Grand Potential of an alloy can be written in terms of (non-)interaction contributions as:

$$\langle \Omega \rangle = F_{non-int} - \langle \Phi_{int} \rangle - \mu N_{atoms}$$

where  $F_{non-int} = -k_B T \sum_{\alpha=1}^N c_{\alpha} \ln c_{\alpha}$

Diffuse scattering experiments on a disordered state reveal the *chemical ordering* fluctuations (or SRO), analogous to “phonon modes”, which are unstable but potentially long-lived.

We study the linear-response to ordering about the (partial) disordered state, since the second-order terms give the SRO. The equations for SRO pair-correlations are EXACT!

$$\alpha_{cpa}^{-1}(\mathbf{q}; T) = \frac{(\frac{\delta_{\alpha\beta}}{c_{\alpha}} - \frac{1}{c_{Host}}) - \beta S_{\alpha\beta}^{(2)}(\mathbf{q}; T)}{c_{\alpha}(\delta_{\alpha\beta} - c_{\beta})} \quad S^{(2)}(\mathbf{q}; T) = F.T \left[ \frac{\delta^2 \langle \Omega \rangle}{\delta c_i \delta c_j} \right]_{c_i}$$

Approximations yield tractable solution, however, but these also lead to errors – in *temperature scale* (say, from using mean-field thermodynamics) or in *electronic energetics* (say, from using single-site, mean-field averaging).

**Exact Electronic DFT Approach**

The Gibbs’ relation of particle number and chemical potential permits, in principle, a means to construct an electronic DFT for the (partially) disordered state. That is, with particle number related to the integrated (DOS) density of state N(E;μ),

$$\langle N(\mu) \rangle = - \frac{\partial \langle \Omega(T, V, \mu) \rangle}{\partial \mu}$$

$$\langle \Omega(T, V, \mu) \rangle = - \int d\mu \langle N(\mu) \rangle = - \int d\mu f(E - \mu) \langle N(E; \mu) \rangle$$

Given an analytic expression of the configurationally average N(E;μ), we may obtain an analytic expression for the grand potential for disordered, partially ordered, or fully ordered (Mermin’s theorem) state.

**1. Improved Electronic Configurational Averaging**

In the *single-site CPA*, we obtain [2]

$$\langle \Omega(T, V, \mu) \rangle_{SS-CPA} = - \int d\mu f(E - \mu) \langle N(E; \mu) \rangle_{SS-CPA}$$

the basis for KKR-CPA total energy calculations in use today, and has been implemented for homogenous disordered case [3] in ternary metallic alloys.

Recent *coarse-graining* concepts developed for *Dynamical Mean-Field Theory* [4] provides a means to go beyond single-site CPA [5] averaging over local, multi-site clusters compatible with point-group symmetry of the underlying Bravais lattice.

$$\langle \Omega(T, V, \mu) \rangle_{nl-cpa} = - \int d\mu f(E - \mu) \langle N(E; \mu) \rangle_{nl-cpa}$$

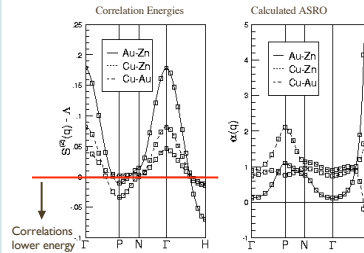
Using a cluster-based average *Non-Local CPA* [5], we have derived an *analytic* expression for the  $\langle N \rangle_{nl-cpa}$ , which requires numerical implementation and testing.

The original theory [5] has been implemented to tested the DOS for 1-D square-well potential, and not the integrated DOS. We are implementing the NL-CPA within our existing KKR-CPA (3-D) code, which will then be a basis for SRO calculations.

**2. Extension to Multisublattice Case**

We have extend the KKR-CPA (single-site) theory for the multicomponent alloy. The above equations generalize with two more superscripts (IJ) that designate the interacting sublattices having species α and β, except that the Brillouin zone is that given by the partially-ordered symmetry.

However, the mean-field thermodynamic approximation is potentially more severe for two-component case. Here we present the calculated SRO and correlation energy given by S<sup>(2)</sup>(q) for fully-disordered bcc Cu<sub>2</sub>AuZn.



**Results**

- Correlation energy leads to SRO peaks at k=(111) or H-point
- Secondary SRO at k=(0.5, 0.5, 0.5) or P-point.
- These indicated high-T B2 and low-T Heusler transitions, as is observed (unpublished).
- Temperature scale is 50% in error.

We have employed the so-called *Onsager corrections* Λ, so intensity is conserved. Such correction provide good temperature scale in binary systems driven by formation energy, the temperature scale (e.g., fcc PdRh we obtained 1080 K and observed is 1050 K).

**Potential Problem**

For a ternary, for example, there are 3 pair-correlations that together must conserve intensity, potentially requiring k-dependent renormalization of intensities to get instability temperatures more correct. Need to improve mean-field thermodynamics to make still tractable with better T scale.

**3. Using Better Thermodynamic Mean-Field Approximations for Improved T.**

We are extending the KKR-CPA-based SRO formula to improve the self-energies correction by summing **all cyclic diagrams** to O(1/Z), where Z is the number of neighbor. These correction maintain required intensity sum rules (as with mean-spherical model and Onsager) but renormalize the pair-correlation in k-dependent manner. Recently these corrections have been called the Ring approximation [7], where they have been tested for lattice-gas and near-neighbor Ising model.

Example of the effect of summing cyclic diagrams [7]:

1-D Ising model (T <sub>c</sub> in units of kT/4J)	exact	MFT	MFT+cyclic
	0.0	1/2	0.22
2-D square lattice Ising model (T <sub>c</sub> in units of kT/4J)	exact	MFT	MFT+cyclic
	<b>0.57</b>	1.0	<b>0.62</b>
3-D fcc Ising model (T <sub>c</sub> in units of kT/4J)	“exact” (MC)	MFT	MFT+cyclic
	<b>2.45</b>	3.0	<b>2.41</b>

We are currently testing a multicomponent version of this approach to assess its validity and usefulness.

**Future**

- Complete and test KKR-NL-CPA for electronic-structure.
- Test “cyclic” corrections for multi-component Ising case.
- Validate the analyticity of the derived analytic expression for the NL-CPA integrated DOS, which is required for thermodynamics (probably by 1-D model Hamiltonian).
- Combine the three for NL-CPA calculations of SRO.
- Address numerical issues required to implement either CPA or NL-CPA into usable and extensible code.

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