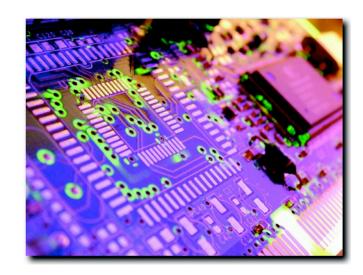
2007 Summer School on Computational Materials Science Quantum Monte Carlo: From Minerals and Materials to Molecules July 9 –19, 2007 • University of Illinois at Urbana–Champaign http://www.mcc.uiuc.edu/summerschool/2007/qmc/

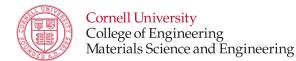
# Electronic Properties of Materials from Quantum Monte Carlo

### **Richard G. Hennig**

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- What are electronic properties?
- Band structure of perfect crystals
- Dopants and defects in semiconductors
- What can we calculate in Quantum Monte Carlo?





## **Electronic Properties of Materials**

#### All materials:

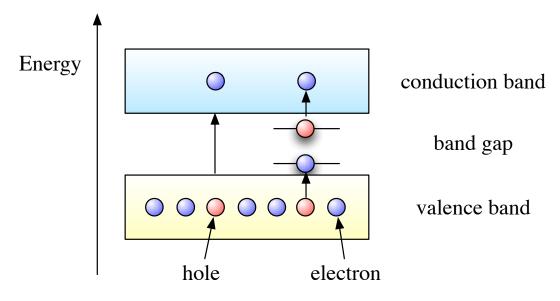
- Electronic density of states and bandgaps
- Electronic and thermal conductivity

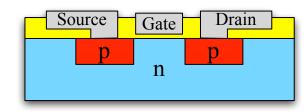
### **Minerals under pressure**

• Metal/insulator transitions affects Earth's magnetic field

#### Semiconductors

- Electron and hole mobility
- Effective masses
- Electronic transition levels of dopants
- Dopant diffusion
- Optical properties, excitons
- Band offsets for heterostructures





### **Band structure of crystals (1)**

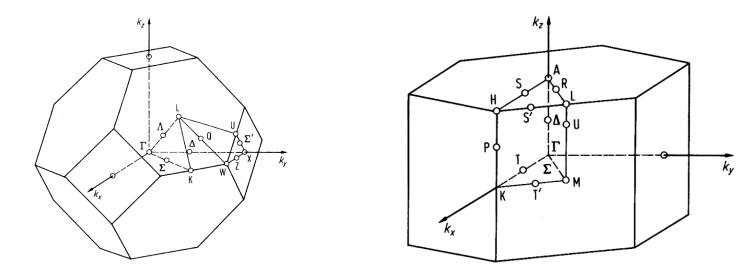
#### **Bloch states and Brilloin zones**

- Crystal structures defined by Bravais lattice  $\{a_i\}$  and basis
- Periodic density  $\Rightarrow$  Bloch theorem  $\psi(\mathbf{r} + \mathbf{R}) = \psi(\mathbf{r}) \cdot \exp(i\mathbf{k} \cdot \mathbf{r})$
- Fourier transformation

• *Reciprocal lattice* 

$$n(\mathbf{r}) = \sum_{\mathbf{G}} n_{\mathbf{G}} \cdot \exp\left(i\mathbf{G} \cdot \mathbf{r}\right)$$
$$\mathbf{b}_{\mathbf{k}} = 2\pi \cdot \frac{\mathbf{a}_{\mathbf{l}} \times \mathbf{b}_{\mathbf{m}}}{\mathbf{b}_{\mathbf{k}} \cdot (\mathbf{b}_{\mathbf{l}} \times \mathbf{b}_{\mathbf{m}})}$$

• *Brillouin zone* is the Wigner-Seitz cell of the reciprocal lattice



### **Band structure of crystals (2)**

#### **Diffraction picture for origin of the energy gap**

- Start with a 1D crystal and consider diffraction of electron wave
  - $n\lambda = 2d \cdot \sin \theta$  with d = a and  $\sin \theta = 1$

$$n\lambda = 2a$$

$$k = \frac{2\pi}{\lambda}$$

$$k = \frac{n\pi}{a}$$

$$\lambda \sim a$$

• Take lowest order (n = 1) and consider incident and reflected electron wave

$$\psi_i = e^{ikx} = e^{i\frac{\pi}{a}\cdot x}$$
 and  $\psi_r = e^{-i\frac{\pi}{a}\cdot x}$ 

• Total wave function for electrons with diffracted wave length

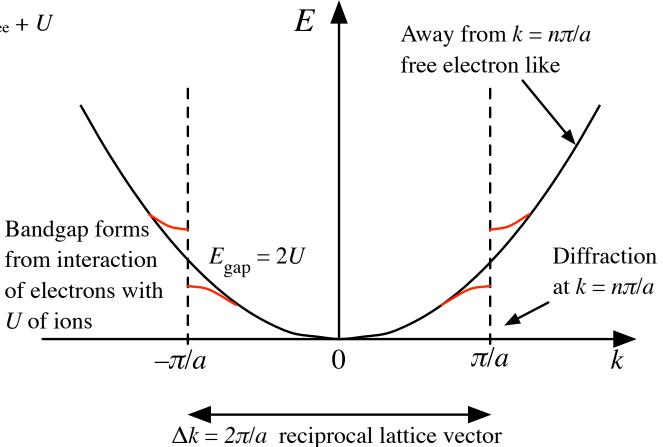
$$\psi = \psi_i \pm \psi_r \quad \Rightarrow \quad \psi_+ = \psi_i + \psi_r = 2\cos\frac{\pi x}{a} \quad \text{and} \quad \psi_- = \psi_i - \psi_r = 2\sin\frac{\pi x}{a}$$

- Only two solutions for  $k = \pi/a$ : Electron density on atoms or between
- No traveling wave solution

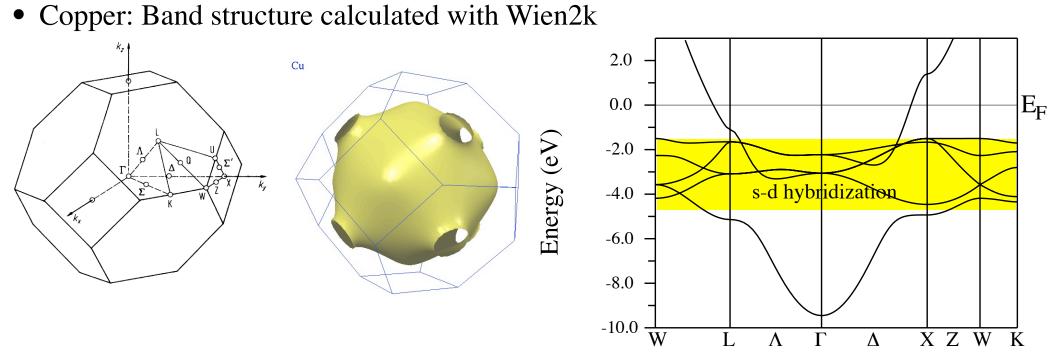
### **Band structure of crystals (3)**

#### Diffraction picture for origin of the energy gap

- If ion potential is a weak perturbation *U*, the electrons near diffraction condition have two possible solutions
  - Electron density between ions:  $E = E_{\text{free}} U$
  - Electron density on ions:  $E = E_{\text{free}} + U$
  - Near diffraction condition energy is parabolic in k, E ∝ k<sup>2</sup>
  - Electron near diffraction conditions are not free
  - Their properties can still be described as "free" with an *effective mass m*\*

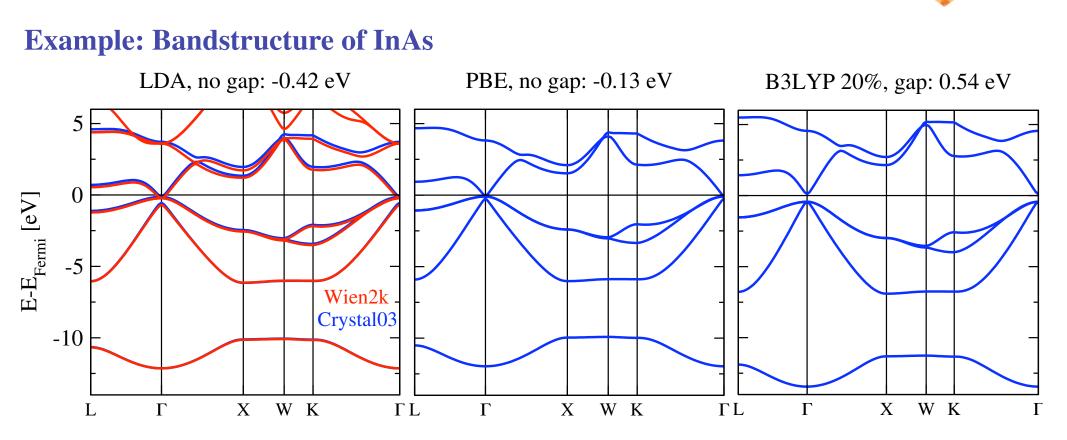


## **Example of metallic band structure: Cu**



- *Nearly free electron s*-band dominates at low and high energies
- Electron near diffraction conditions have different effective mass
- *Hybridization* between *nearly-free s* and *atomic-like d* orbitals at intermediate energies
- Necking of Fermi surface in [111] directions  $\Rightarrow$  *Hume-Rothery stabilization*

## The bandgap problem of DFT



• Experimental bandgap: 0.41 eV

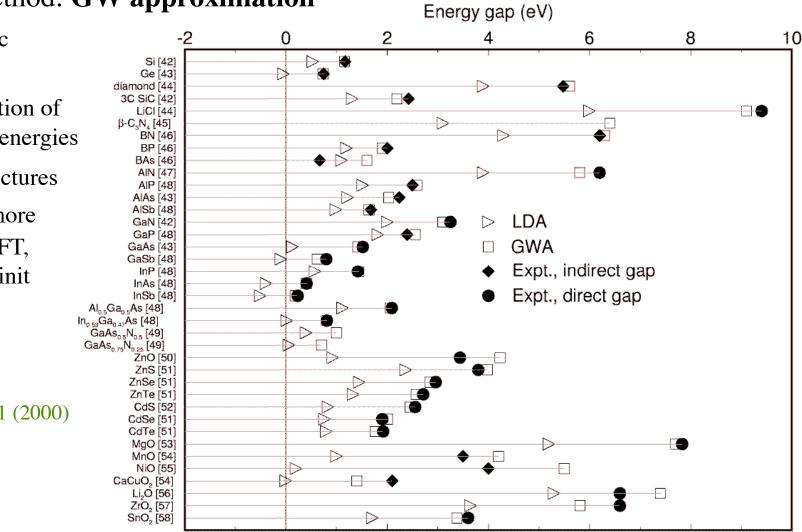
Band gap problem: LDA and GGA yield a metallic ground state!

- Practical solution: **Hybrid functionals** B3LYP & HSE (0.39 eV)
- Better solution: **GW approximation or QMC methods**

## **GWA calculation of band structures**

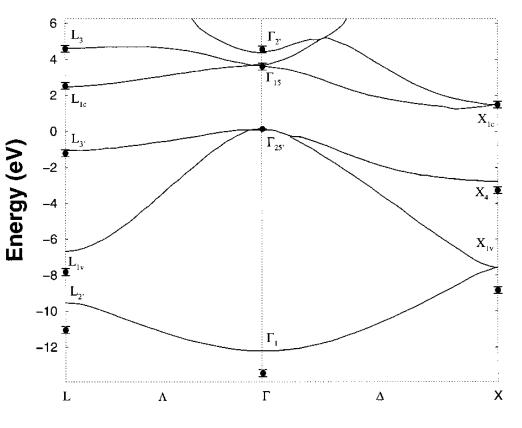
- Density functional methods provide a fast way of getting band structures
- However many functionals suffer from the **band gap problem**
- More accurate method: GW approximation
  - Based on electronic Green's function
  - Many-body correction of DFT quasiparticle energies
  - Accurate band structures
  - Computationally more demanding than DFT, implemented in abinit

Aulbur *et al*. Solid State Phys. 54, 1 (2000)



## **Calculation of band structures (QMC)**

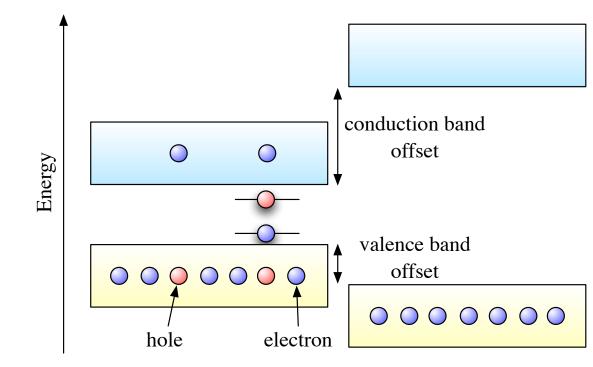
- Quantum Monte Carlo
  - Variational for ground state of respective symmetry
  - Combined optimization of ground and excited states to keep wave functions orthogonal (Schautz *et al.*, J. Chem. Phys. 121, 5836)
  - Silicon band structure shown
- Comparison QMC vs. GWA
  - Similar accuracy
  - Both computationally expensive
  - QMC much less tested than GWA
- For molecular systems, quantum chemistry methods such as MCSCF are very powerful



Williamson et al. PRB 57, 12140 (1998)

## **Electronic Properties of Semiconductors**

- Density of states and bandgaps
- Intrinsic and extrinsic semiconductors (p and n doping)
- Heterostructures
  - Fabrication by MBE or MOCVD
  - Electronic properties controlled by band offsets
  - Examples:
    - (1) Laser diodes from II-VI and III-V heterostructures
    - (2) Heterojunction bipolar and high electron mobility transistor
- Calculation of defect levels and band offsets
  - Similar to calculation for crystal
  - DFT bandgap problem  $\Rightarrow$  Use GWA or QMC methods for improved accuracy

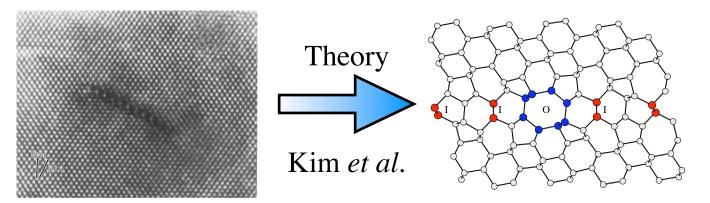


## **Dopants and Defects in Semiconductors**

- Where are they?
  - ▶ Small concentrations and small sizes of dopants and defects
     ⇒ Experimental observations difficult
- What are they doing?
  - Dopants and defects can lead to electronic levels in the band gap
  - n-type donor states (P, As, Sb in Si)
  - p-type acceptor states (B in Si)

Computational methods can link experimental electronic properties to dopant and impurity structures

{311} defect
in B doped Si



#### **Harmonic transitions state theory**

• Calculation of diffusion constant and reaction rates

$$D_{i} = \frac{a_{i}^{2}}{6} \cdot \beta_{i} \cdot f_{i} \cdot \Gamma_{i}$$

$$a_{i} - \text{Jump length}$$

$$\beta_{i} - \text{Availability factor (i.e. vacancy concentration)}$$

$$\Gamma_{i} = \Gamma_{0} \cdot \exp\left(-\frac{\Delta H}{kT}\right)$$

$$f_{i} - \text{Correlation factor (due to back jumps)}$$

$$\Gamma_{i} - \text{Jump frequency}$$

$$\Delta H - \text{Enthalpy barrier of saddle point}$$

$$\Gamma_{0} = \frac{\prod_{i=1}^{3N-3} \nu_{i}}{\prod_{i=1}^{3N-4} \nu_{i}'}$$

$$\nu_{i} - \text{Phonon frequency at minimum}$$

$$\nu_{i}' - \text{Phonon frequency at saddle point}$$

- Jump length determined by geometry
- Availability factor determined by concentration of available sites
- Correlation factor requires at least forward and backward rate
- How to calculate phonon frequencies (see talk and lab by Dario Alfe)
- How to determine the saddle point structure?

## **Structure of Defects and Dopants**

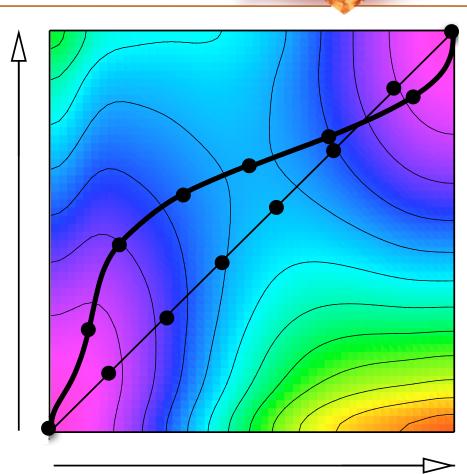
### **Determining saddle points**

- Nudged elastic band method (Jonsson et al. 1998)
  - Chain of 3N-dim configurations connected by "springs"
  - Relaxation  $\Rightarrow$  Minimum energy path
  - Implemented in PWSCF

#### • Dimer method

(Henkelman and Jonsson, JCP (1999))

- Optimize rotation angle of the vector between pair of configurations ⇒ lowest curvature mode
- Follow direction uphill minimizing all other directions
- Very efficient first-derivative-only saddle search method



## **Diffusion of silicon interstitials**

- Diffusion path only known for single but not larger interstitials
- Experiment: Diffusion activation 4.7–5.0 eV, barrier 0.3–1.8 eV
- Energy units: Room temperature = 0.024 eV

### Multiscale approach: DFT & TB

DFT-GGA	$I_1$	$I_2$	$I_3$	$\mathbf{I}_{\mathbf{n}}^{\mathrm{chain}}$
Formation [eV/atom]	3.8	2.8	2.4	1.7
Barrier [eV]	0.3	0.3	0.5	

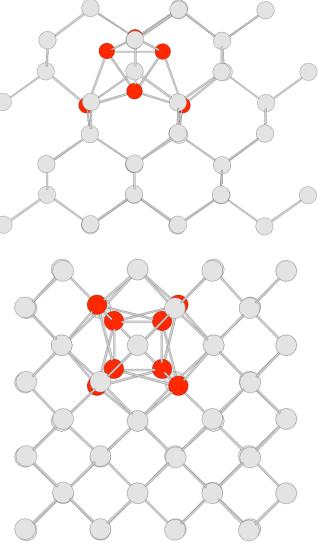
### **Results:**

Fast diffusion of interstitials

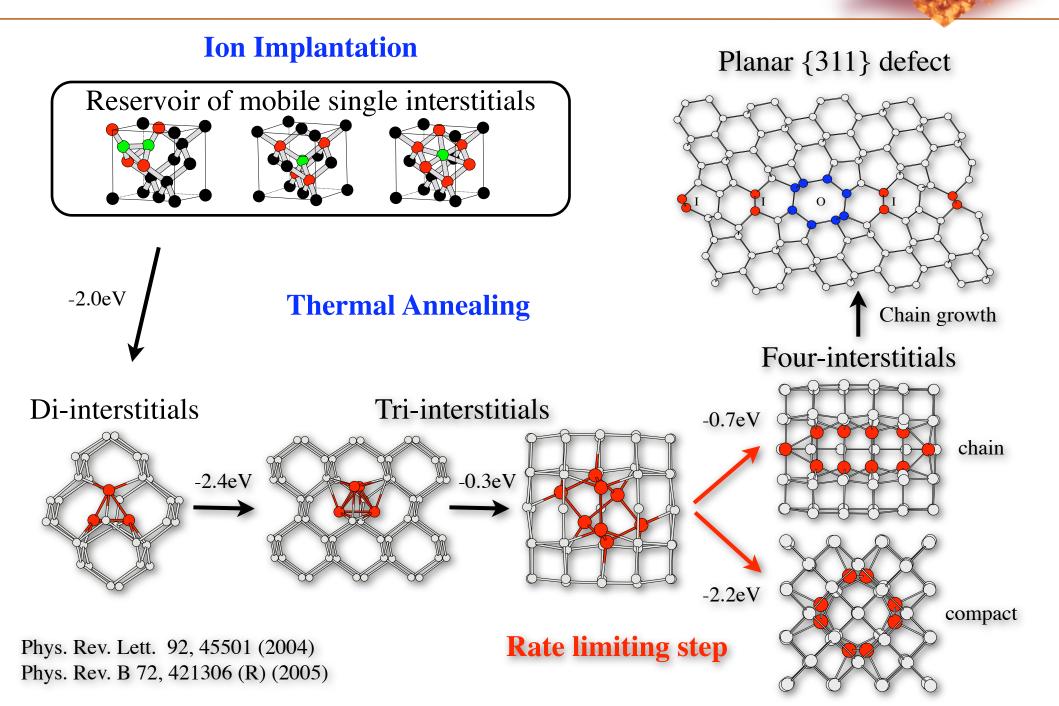
Single interstitials dominate

Driving force to form defect precipitates

Defect charge states can lower diffusion barrier



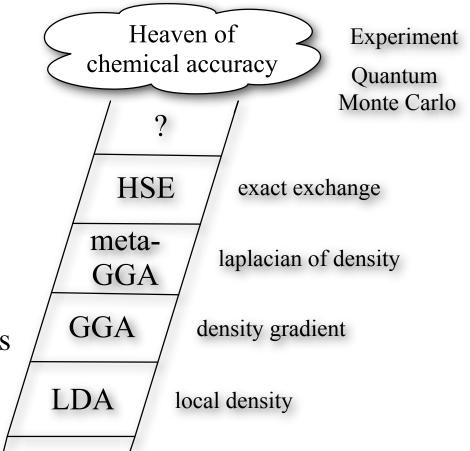
### From compact to extended defect structures



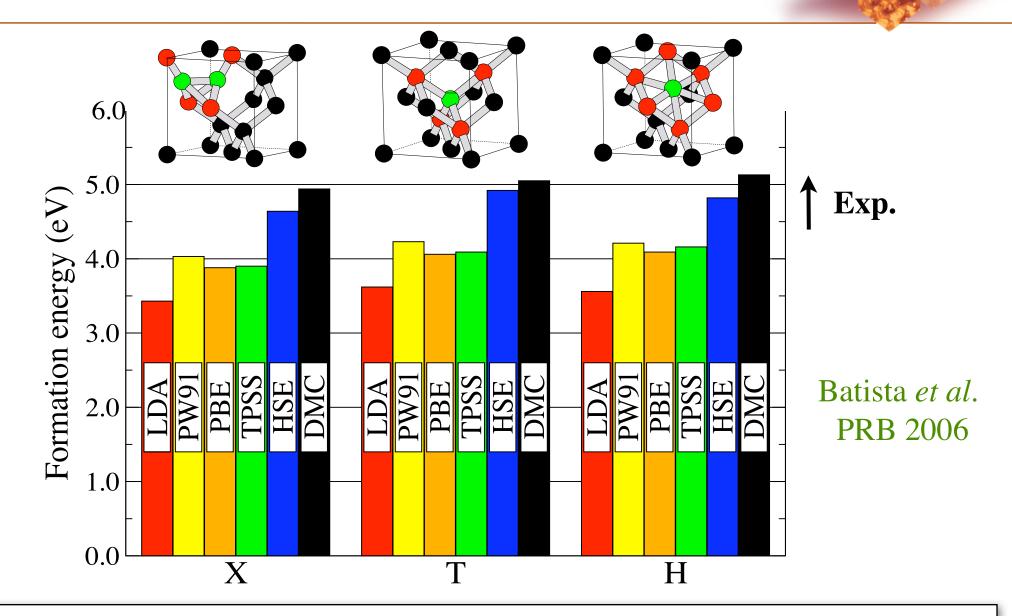
## **Jacobs ladder of density functionals**

### **Approximations for unknown density functional**

- Climbing "Jacob's ladder" to heaven of chemical accuracy (Perdew *et al.* PRL 2003)
- Comparison to experiment or quantum chemistry
- Difficulties:
  - Experimental energies of defects
  - Quantum chemistry methods for solids
- Benchmark calculations by quantum Monte Carlo

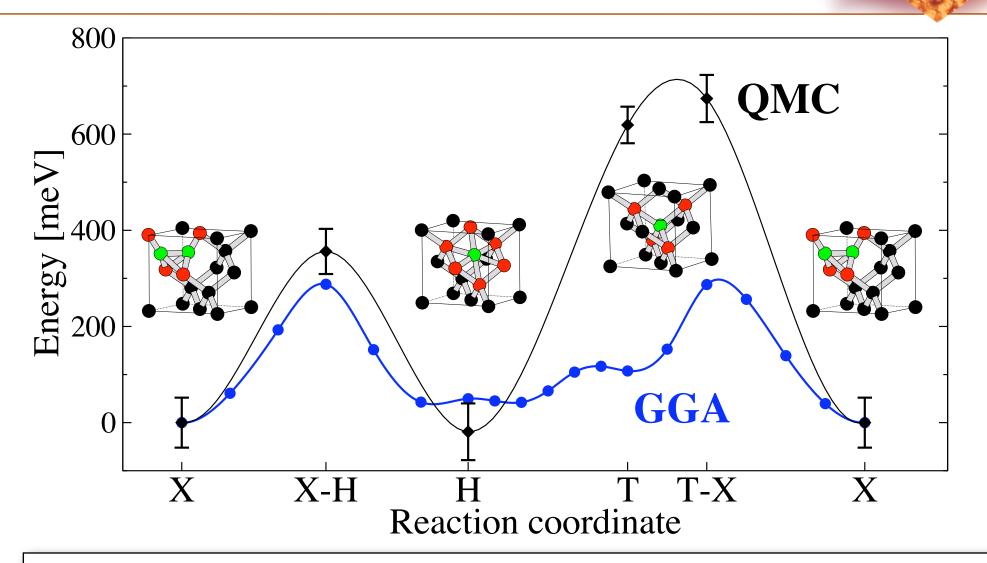


### **Accuracy of defect energies**



Climbing "Jacob's ladder" of density functionals improves the accuracy for defect formation energies. The highest rung-hybrids-agree with QMC.

### **Accuracy of diffusion barriers**



Lowest energy barrier from X to H defect is similar in QMC and DFT. The T defect and its barrier are higher in QMC.

### **Electronic Transition Levels of Dopants**

• Formation energy of defect *X* with charge *q* 

$$E_f[X_q] = E_{\text{tot}}[X_q] - E_{\text{tot}}[\text{bulk}] - \sum_i n_i \cdot \mu_i + q[E_f + E_v + \Delta V]$$

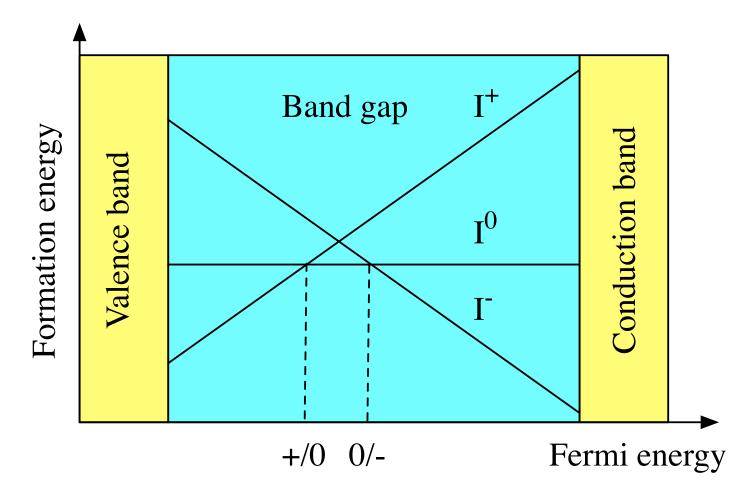
$E_{tot}[X_q]$	Energy for	or charged	defect	(using	uniform	background	charge)

#### *E*<sub>tot</sub>[bulk] Energy of ideal crystal

- $n_i, \mu_i$  Number of defect atoms and their chemical potential
- $E_f$  Fermi energy relative to reference (valence band maximum)
- $E_{v}$  Energy of reference
- $\Delta V$  Alignment of electrostatic potentials of defect and crystal cell

### **Electronic Transition Levels of Dopants**

- Charge of defects changes as a function of Fermi level
- Change of defect charge for increasing Fermi level ++/+ +/0 0/- -/--
- *Thermodynamic* transition levels: Include relaxations of final state
- Optical transition levels: Final and initial charge state for same geometry



### **Summary: What can we calculate with which method?**

Property	DFT	GWA	QMC
Band gap	not always accurate <sup>1</sup>	very accurate	accurate
Effective mass	yes	yes	no
Transition levels	yes	very accurate	not done
Band offsets	not always accurate <sup>1</sup>	very accurate	not done
Defect energies	not always accurate <sup>1</sup>	no	accurate
Barriers	not always accurate <sup>1</sup>	no	accurate

<sup>1</sup> Improved accuracy for hybrid functionals



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